

METALLURGICAL ASPECTS OF THE SOLUBILIZATION OF MICROALLOYING ELEMENTS DURING STEEL SLAB REHEATING

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INTRODUCTION



- Microalloyed steels must be reheated before hot rolling to a **minimum temperature** that ensures full Nb solubilization – this is a vital requisite for a **successful controlled rolling**.
- There is several equations available for the calculation of this minimum temperature, but the **minimum time** under this temperature that is needed for full Nb solution is generally not known.
- **The detection of Nb precipitates** in microstructure is not exactly easy, requiring transmission electronic microscopy.

INTRODUCTION



- Under practical conditions it is assumed that the full Nb solution demands at least **from 10 to 30 minutes** under temperatures above those calculated by the equilibrium equations.
- However, the demand for minimum dispersion in the final mechanical properties of rolled products requires **a better knowledge about the kinetics of the solubilization** of microalloying elements.
- An industrial continuous cast slab of microalloyed steel presents microalloy precipitates with **several shapes and sizes across its thickness**.

INTRODUCTION

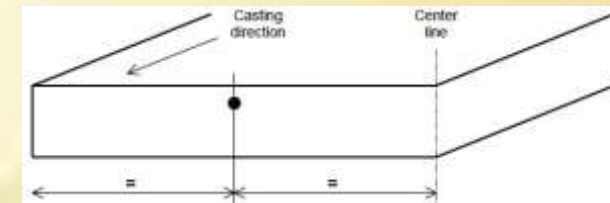
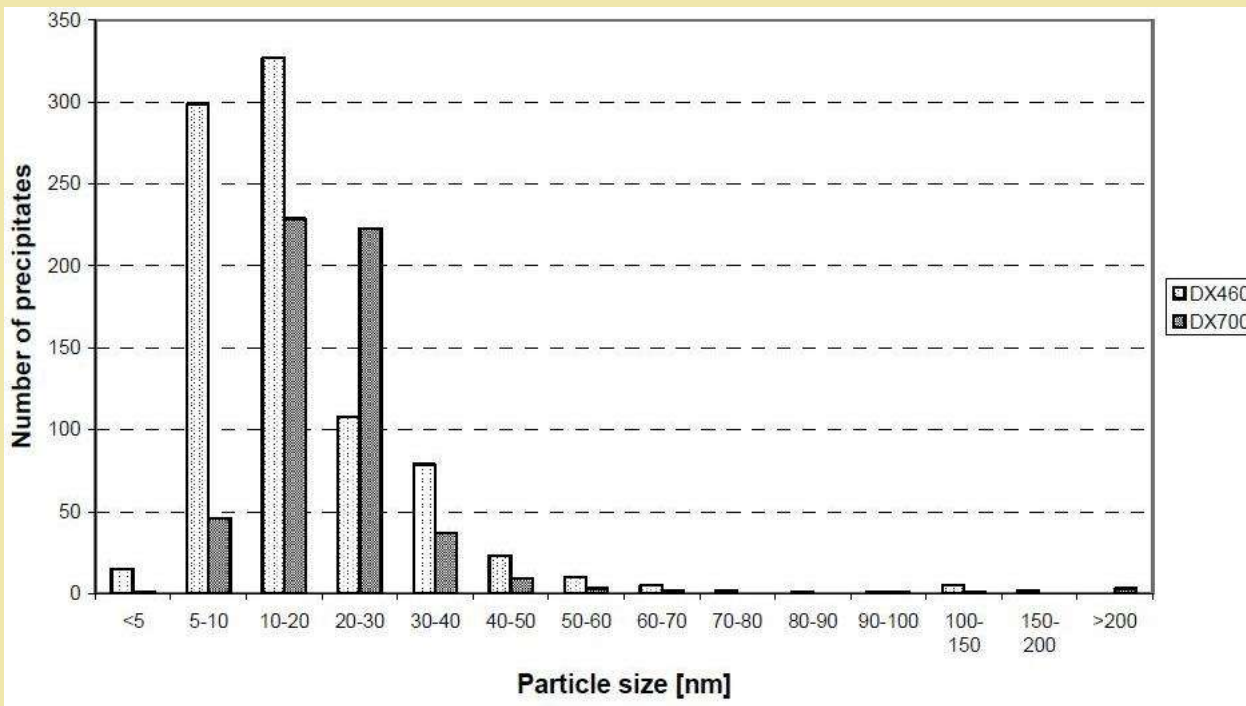


- **Typical slab size distribution** of microalloy precipitates in a NbTiV structural steel:
 - 80%: spherical, 1 to 50 nm;
 - 10%: cubical, 60 to 300 nm;
 - 10%: star/wing, 150 to 600 nm.
- Partition of Nb during solidification is very high: the interdendritic liquid steel has Nb content **7 times higher** than the nominal chemical composition.
- No wonder that the interdendritic region of the slab shows a **higher precipitate concentration**.

INTRODUCTION

- **Size precipitate distribution** in microalloyed steel slabs:

| Steel | C | Si | Mn | N | Mo | Al | Nb | Ti | V |
|----------|-------|-------|------|--------|-------|-------|-------|-------|-------|
| Domex700 | 0.061 | 0.082 | 1.87 | 0.0084 | 0.12 | 0.041 | 0.061 | 0.099 | 0.007 |
| Domex460 | 0.064 | 0.058 | 1.38 | 0.0048 | 0.002 | 0.037 | 0.048 | 0.004 | 0.005 |



INTRODUCTION



- A microstructural analysis of commercial hot coils of microalloyed steels showed that **up to 49% of the microalloy contents are lost** (not solubilized during reheating) as they are present as **coarse eutectic precipitates** (size greater than **500 nm**):

| Aço | C | Mn | Si | P | S | Al | Nb | Ti | V | N |
|--------|------|------|------|-------|-------|------|------|------|------|--------|
| NbTi-1 | 0,05 | 0,55 | 0,02 | n,d, | n,d, | 0,02 | 0,02 | 0,06 | - | 0,006 |
| NbTi-3 | 0,11 | 1,54 | 0,28 | 0,026 | 0,007 | 0,01 | 0,04 | 0,11 | - | 0,008* |
| NbTiV | 0,14 | 1,38 | 0,25 | 0,018 | 0,007 | 0,07 | 0,04 | 0,04 | 0,03 | 0,008 |

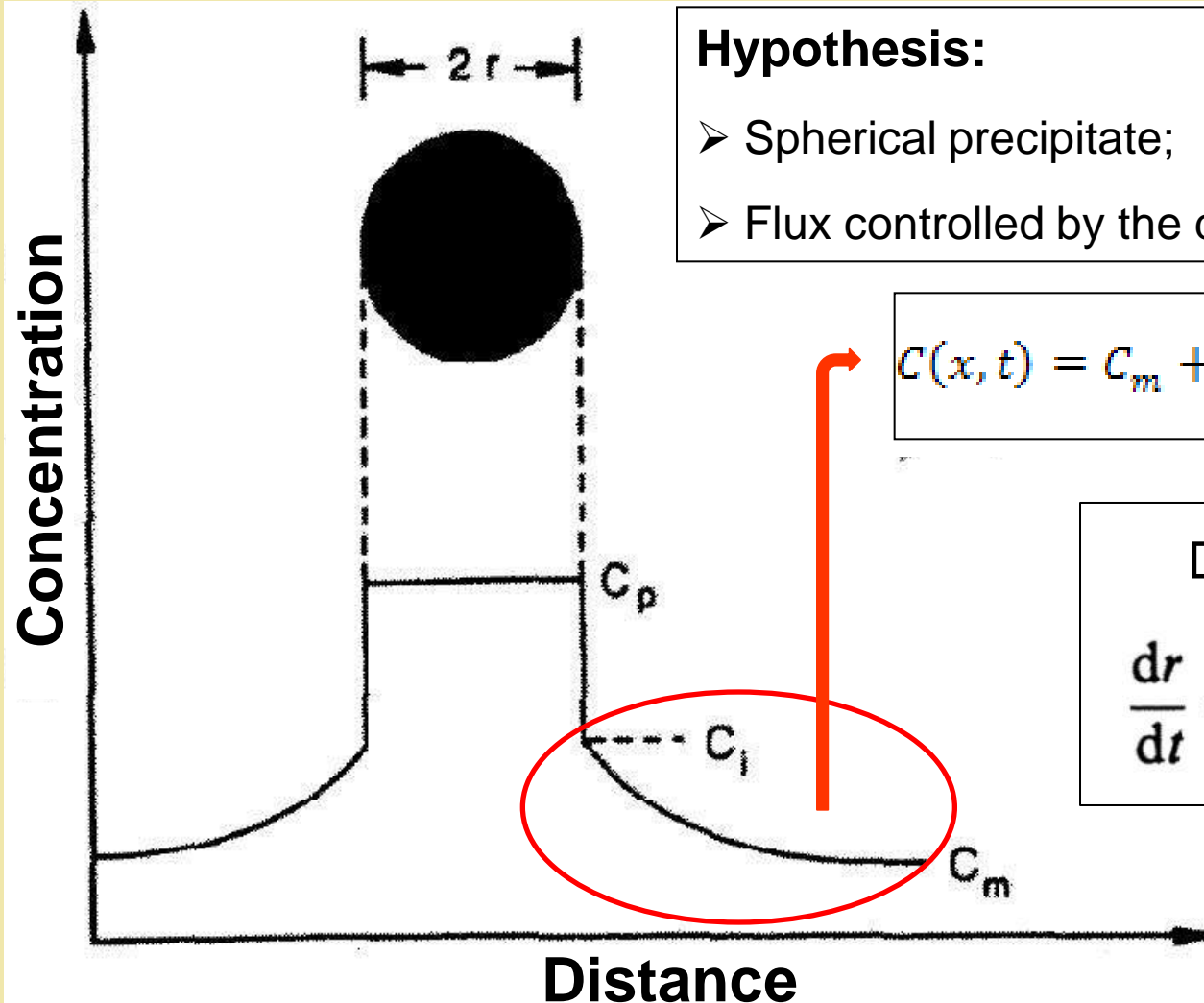
| Aço | $V_{f\max}$ teórico | carbonitretos eutéticos | carbonitretos insolúveis | Perda total de microligantes | TiN_{\max} teórico | perda adicional devido segregação |
|--------|-----------------------|-------------------------|--------------------------|------------------------------|-----------------------|-----------------------------------|
| NbTi-1 | 13.1×10^{-4} | 5.6×10^{-4} | 0.8×10^{-4} | 49 % | 4.01×10^{-4} | 60 % |
| NbTi-3 | 26.1×10^{-4} | 8.5×10^{-4} | desconhecido | > 33 % | 5.35×10^{-4} | > 59 % |
| NbTiV | 17.4×10^{-4} | 5.0×10^{-4} | 1.6×10^{-4} | 38 % | 5.35×10^{-4} | 23 % |

INTRODUCTION



- All these facts led to the development of this work, whose aims were:
 - The development of a mathematical model for the calculation of the **solubilization kinetics of microalloy precipitates** under the typical industrial slab reheating conditions;
 - A better knowledge about the possible effects of variations in the **precipitate size, chemical composition and temperature** in the austenite solubilization of microalloyed steels.

DISSOLUTION



Hypothesis:

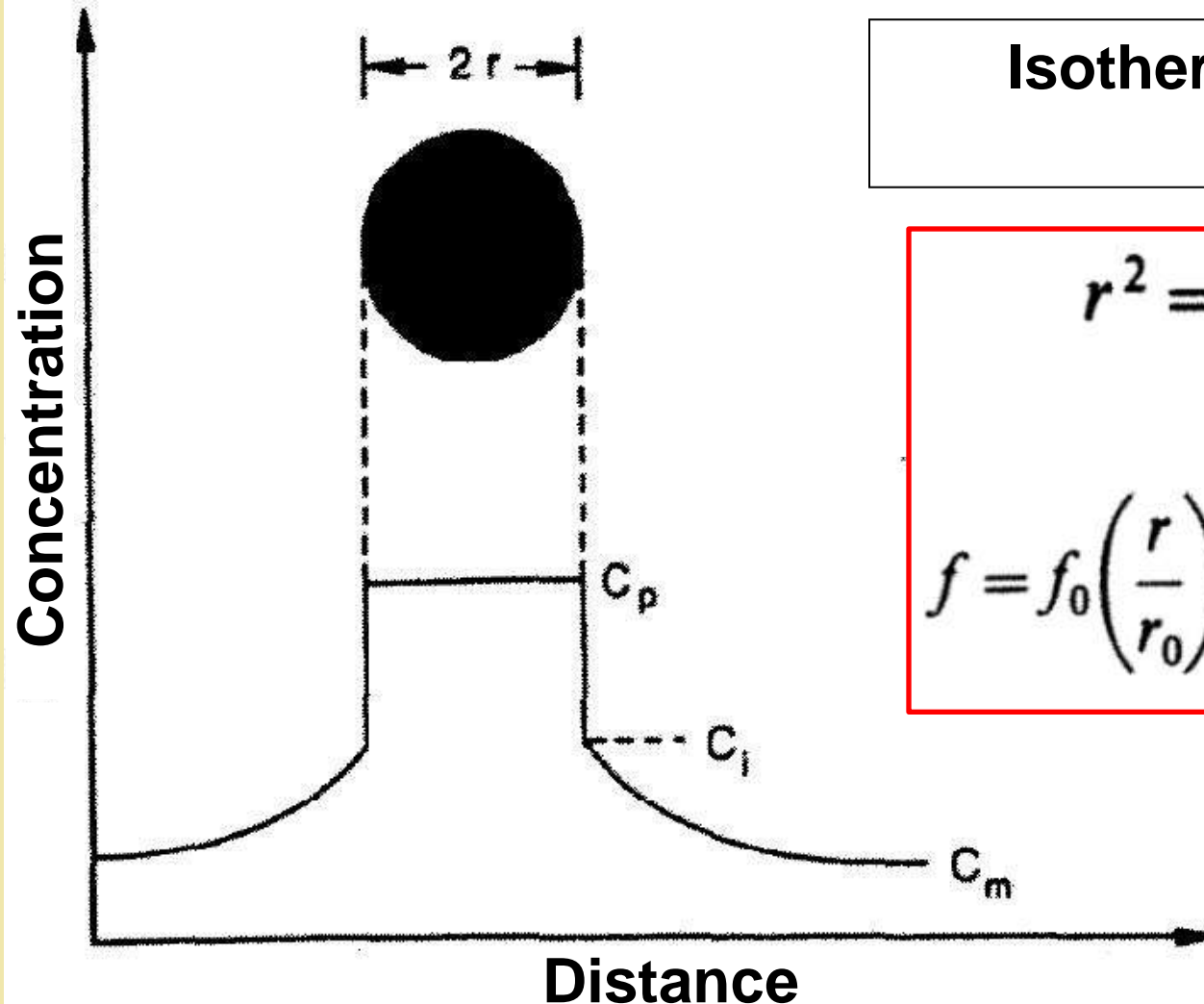
- Spherical precipitate;
- Flux controlled by the diffusion of the slowest atom.

$$C(x, t) = C_m + (C_p - C_m) \operatorname{erfc}\left(\frac{x - r}{2\sqrt{D_m t}}\right)$$

Dissolution Rate:

$$\frac{dr}{dt} = \alpha \left[\frac{D_m}{r} - \sqrt{\frac{D_m}{\pi t}} \right]$$

ISOTHERMAL DISSOLUTION



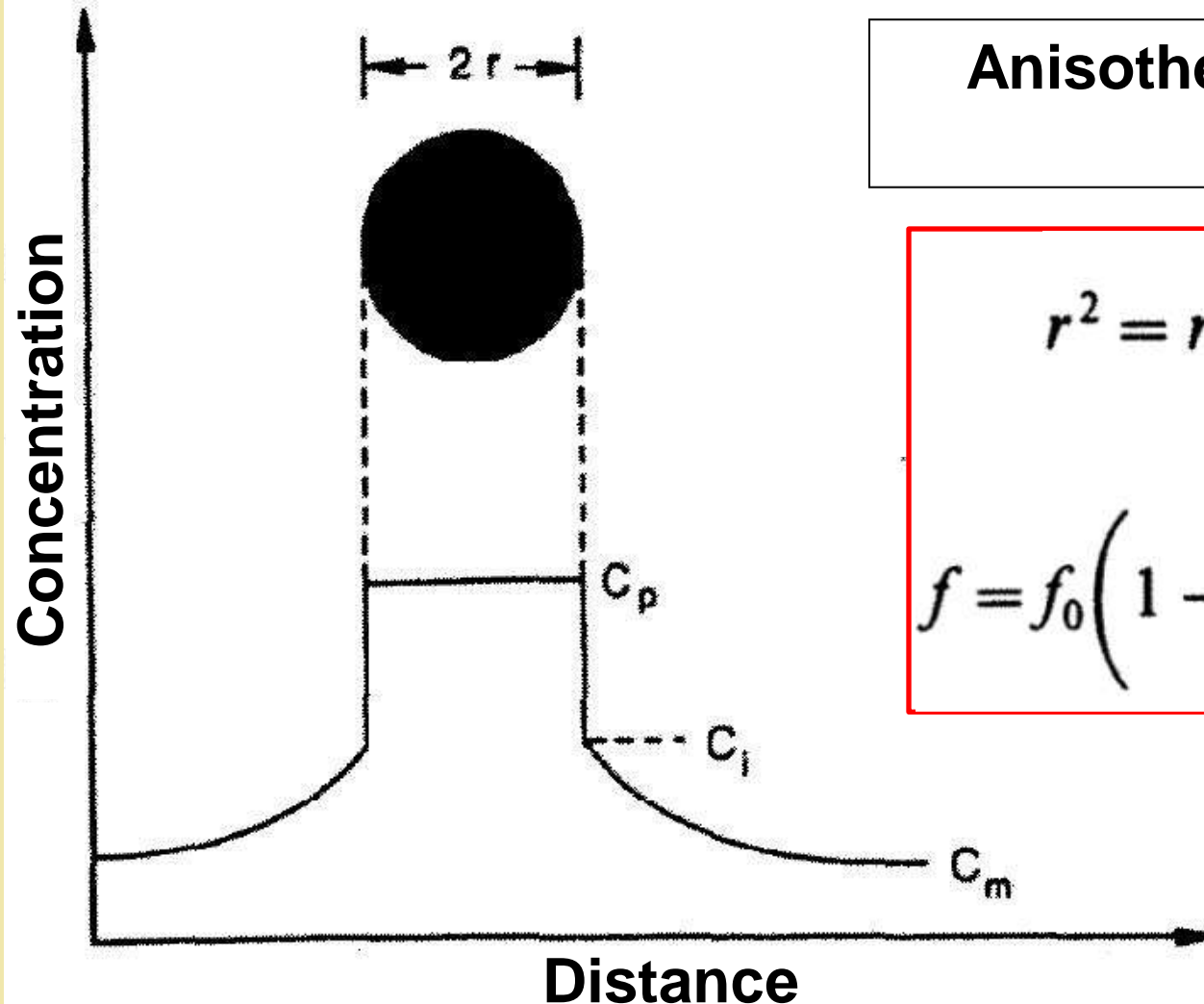
Isothermal Dissolution Kinetics:

$$r^2 = r_0^2 - 2\alpha D_m t$$

$$f = f_0 \left(\frac{r}{r_0} \right)^3 = f_0 \left(1 - \frac{2\alpha D_m t}{(r_0)^2} \right)^{3/2}$$

$$\alpha = \frac{C_i - C_m}{C_p - C_i}$$

ANISOTHERMAL DISSOLUTION



Anisothermal Dissolution Kinetics:

$$r^2 = r_0^2 - 2 \int_{t_1}^{t_2} \alpha D_m dt$$

$$f = f_0 \left(1 - \frac{2}{(r_0)^2} \int_{t_1}^{t_2} \alpha D_m dt \right)^{3/2}$$

$$\alpha = \frac{C_i - C_m}{C_p - C_i}$$

ANISOTHERMAL DISSOLUTION



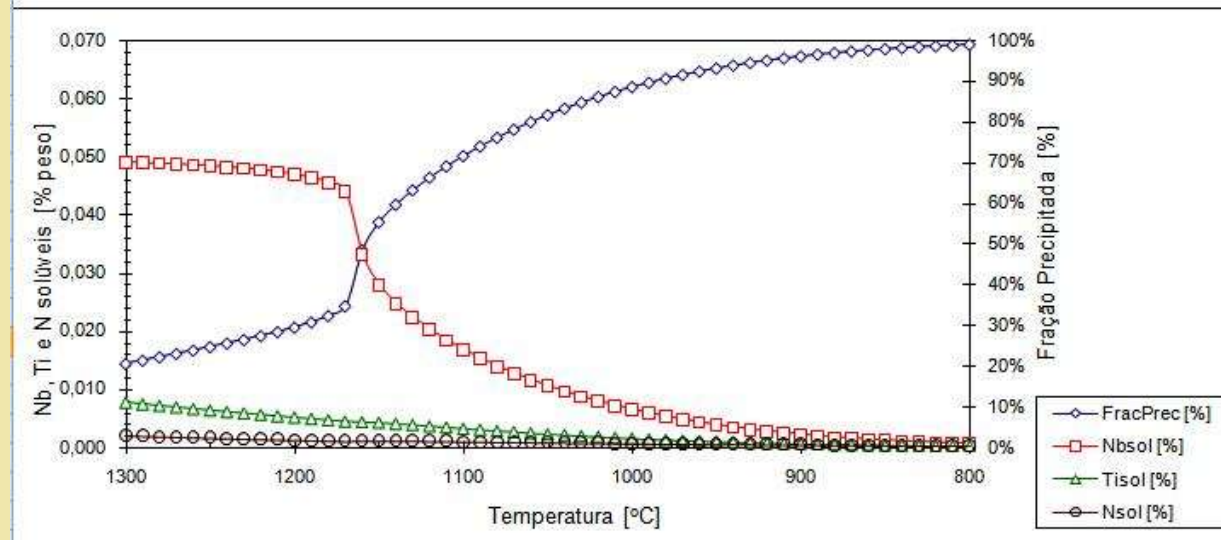
- The anisothermal dissolution implies in an **additional complication**, as the precipitate fraction and its chemical composition (and, consequently, the composition of the matrix) are function of the **temperature**.
- The solution of this problem requires the use of a mathematical model to calculate the **thermodynamical equilibrium between precipitates and austenite**, which was already developed.

AUSTENITE EQUILIBRIUM

| | | | | | |
|--------------------------------|-------------------|------|-------|-------|--------|
| Amostra | Aço NbTi Genérico | | | | |
| Composição Química [% peso] | C | Mn | Nb | Ti | N |
| | 0,11 | 1,53 | 0,050 | 0,016 | 0,0043 |

11/03/2009
15:40

| T [°C] | f | FracPrec [%] | Carbonitreto Nb _x Ti _(1-x) C _y N _(1-y) | | | | Nb _{sol} [%] | Ti _{sol} [%] | C _{sol} [%] | N _{sol} [%] |
|-----------|---------|-----------------|--|-------|------|-------|--------------------------|--------------------------|-------------------------|-------------------------|
| | | | x | (1-x) | y | (1-y) | | | | |
| 1300 | 0,00020 | 21% | 0,05 | 0,95 | 0,08 | 0,92 | 0,0490 | 0,0078 | 0,1094 | 0,0020 |
| 1250 | 0,00024 | 25% | 0,07 | 0,93 | 0,10 | 0,90 | 0,0484 | 0,0064 | 0,1093 | 0,0016 |
| 1200 | 0,00029 | 30% | 0,12 | 0,88 | 0,15 | 0,85 | 0,0469 | 0,0051 | 0,1091 | 0,0012 |
| 1150 | 0,00054 | 55% | 0,49 | 0,51 | 0,53 | 0,47 | 0,0278 | 0,0042 | 0,1065 | 0,0011 |
| 1100 | 0,00069 | 72% | 0,57 | 0,43 | 0,62 | 0,38 | 0,0167 | 0,0032 | 0,1050 | 0,0010 |
| 1050 | 0,00079 | 82% | 0,60 | 0,40 | 0,65 | 0,35 | 0,0105 | 0,0022 | 0,1041 | 0,0008 |
| 1000 | 0,00086 | 88% | 0,61 | 0,39 | 0,66 | 0,34 | 0,0064 | 0,0015 | 0,1035 | 0,0007 |
| 950 | 0,00090 | 93% | 0,61 | 0,39 | 0,67 | 0,33 | 0,0038 | 0,0009 | 0,1031 | 0,0006 |
| 900 | 0,00093 | 96% | 0,61 | 0,39 | 0,67 | 0,33 | 0,0022 | 0,0005 | 0,1029 | 0,0004 |
| 850 | 0,00095 | 98% | 0,62 | 0,38 | 0,67 | 0,33 | 0,0012 | 0,0003 | 0,1028 | 0,0004 |
| 800 | 0,00096 | 99% | 0,62 | 0,38 | 0,67 | 0,33 | 0,0006 | 0,0002 | 0,1027 | 0,0003 |



MODEL APPLICATIONS



- Dissolution kinetic studies generally have a **microstructural approach**, analysing precipitate diameter and fraction evolution along time.
- In this work dissolution was expressed in terms of **dissolved Nb in austenite**, a parameter much more relevant for industrial objectives.
- The heating curve used as input for the model was measured under **industrial conditions**, using a thermocouple placed at **half slab thickness**, just above the water cooled skid/slab interface. This point has the **worst thermal soaking condition**.

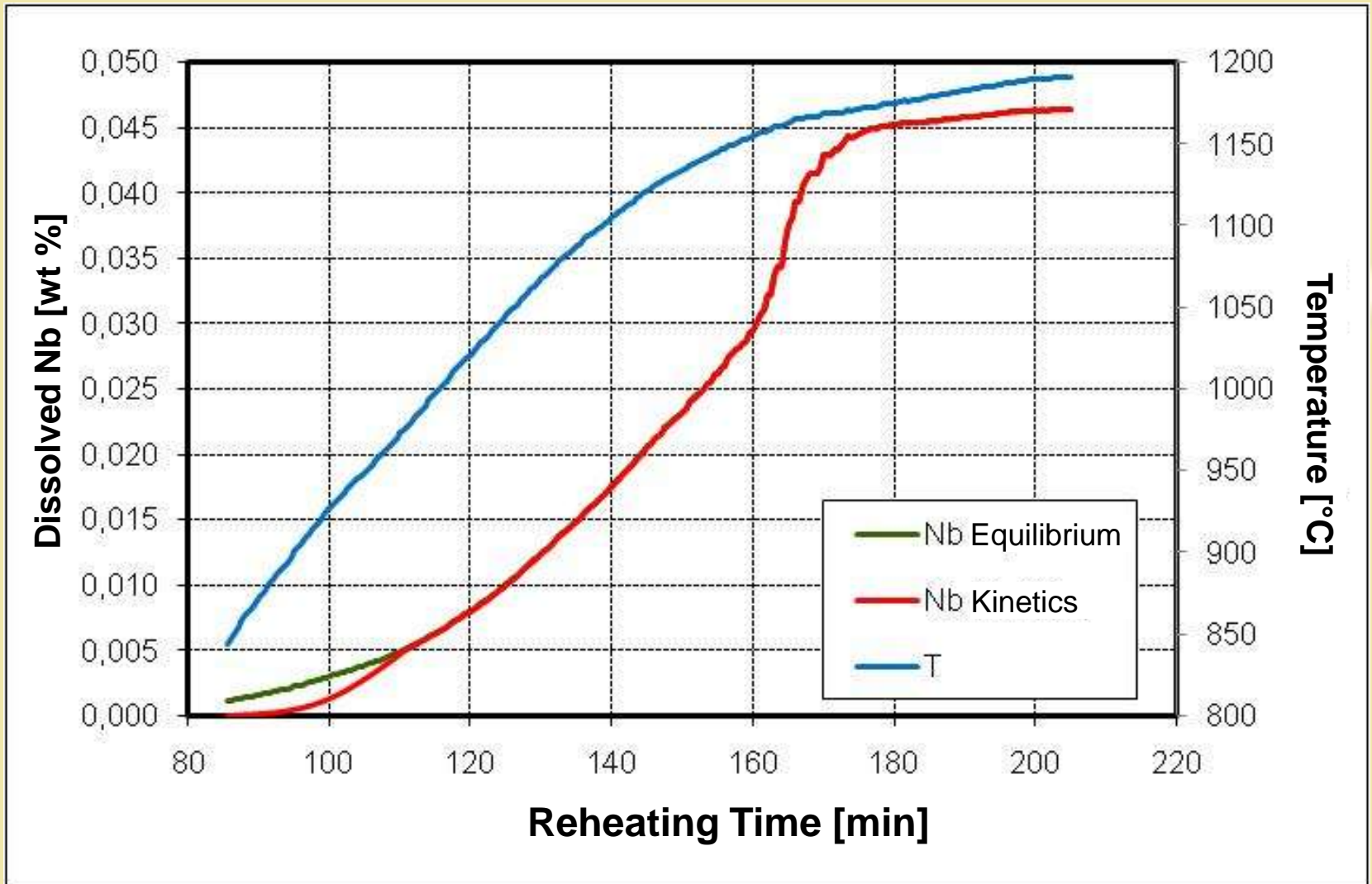
#1: PRECIPITATE SIZE



- Input data:
 - **Chosen steel chemical composition:**
 - 0.11% C
 - 0.050% Nb
 - 0.016% Ti
 - 0.0043% N
 - **Precipitate diameters:**
 - **Average:** 17 nm (according to Borggren, NbTi steel)
 - **Eutectoid:** 250 nm and 500 nm.

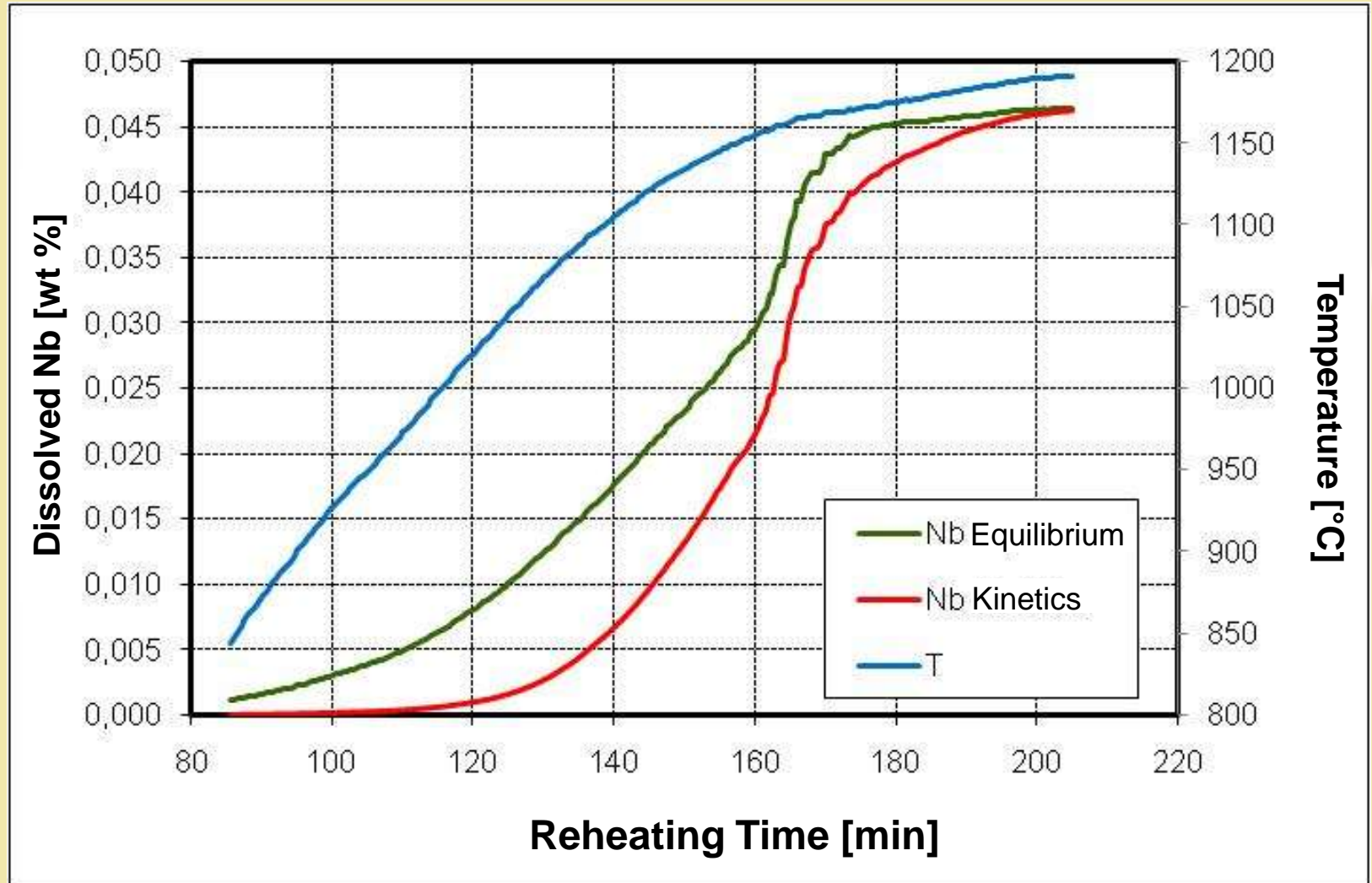
#1: PRECIPITATE SIZE

∅ precipitate: 17 nm



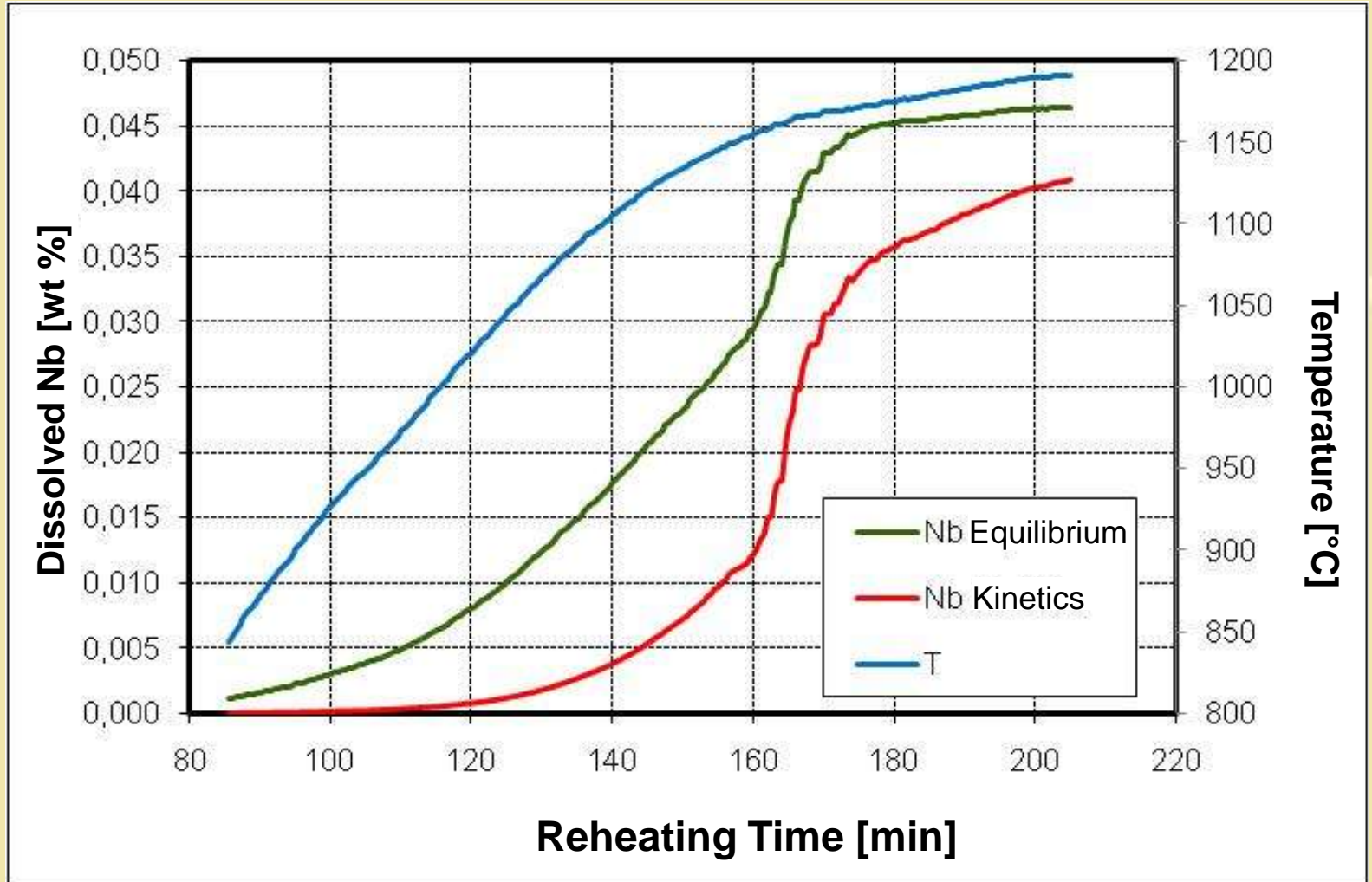
#1: PRECIPITATE SIZE

∅ precipitate: 250 nm



#1: PRECIPITATE SIZE

∅ precipitate: 500 nm



#1: PRECIPITATE SIZE



- For this specific alloy, and considering the worst slab reheating condition, **full Nb solubilization** is achieved for precipitates with a **maximum diameter of 250 nm** – well above the mean precipitate diameter.
- As a matter of fact, models Structura (IRSID) and TACSI (ArcelorMittal) **disregard kinetic effects** during the calculation of microalloy solubilization of during slab reheating – both consider enough only the use of a **thermodynamical equilibrium model**.

Piette, Materials Science Forum, 284-286, 1998, 361-368

Huin, 3rd Int. Conf. On Thermomechanical Processing of Steels - TMP 2008, Padua, 2008

#2: ALLOY COMPOSITION



- All industrial steel alloys are specified within a **chemical composition tolerance range** due to the normal precision limitations of the elaboration and refining processes.
- The model developed here was used to verify **the effect of such tolerance range** in the Nb solubilization evolution.
- **Base alloy:**
 - 0.11% C
 - 0.050% Nb
 - 0.016% Ti
 - 0.0043% Nb

#2: ALLOY COMPOSITION

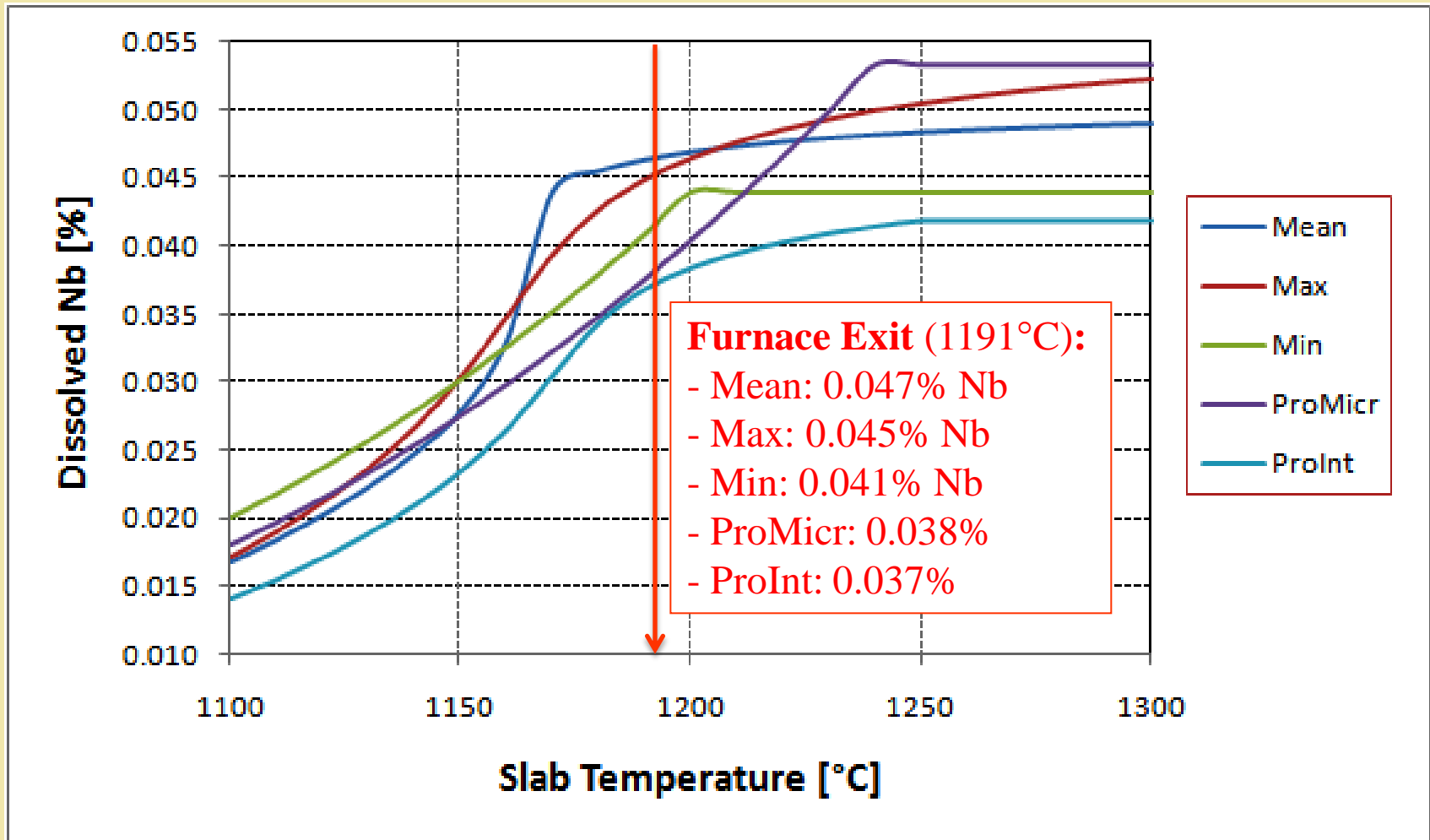


- Definition of **limit compositions** within the normal specification range of an hypothetical microalloyed steel:

| Alloy | C | Nb | Ti | N |
|----------------|----------|-----------|-----------|----------|
| Mean | 0.11 | 0.050 | 0.016 | 0.0043 |
| Min | 0.09 | 0.045 | 0.010 | 0.0005 |
| Max | 0.12 | 0.055 | 0.022 | 0.0080 |
| ProMicr | 0.09 | 0.055 | 0.022 | 0.0005 |
| ProInt | 0.12 | 0.045 | 0.010 | 0.0080 |

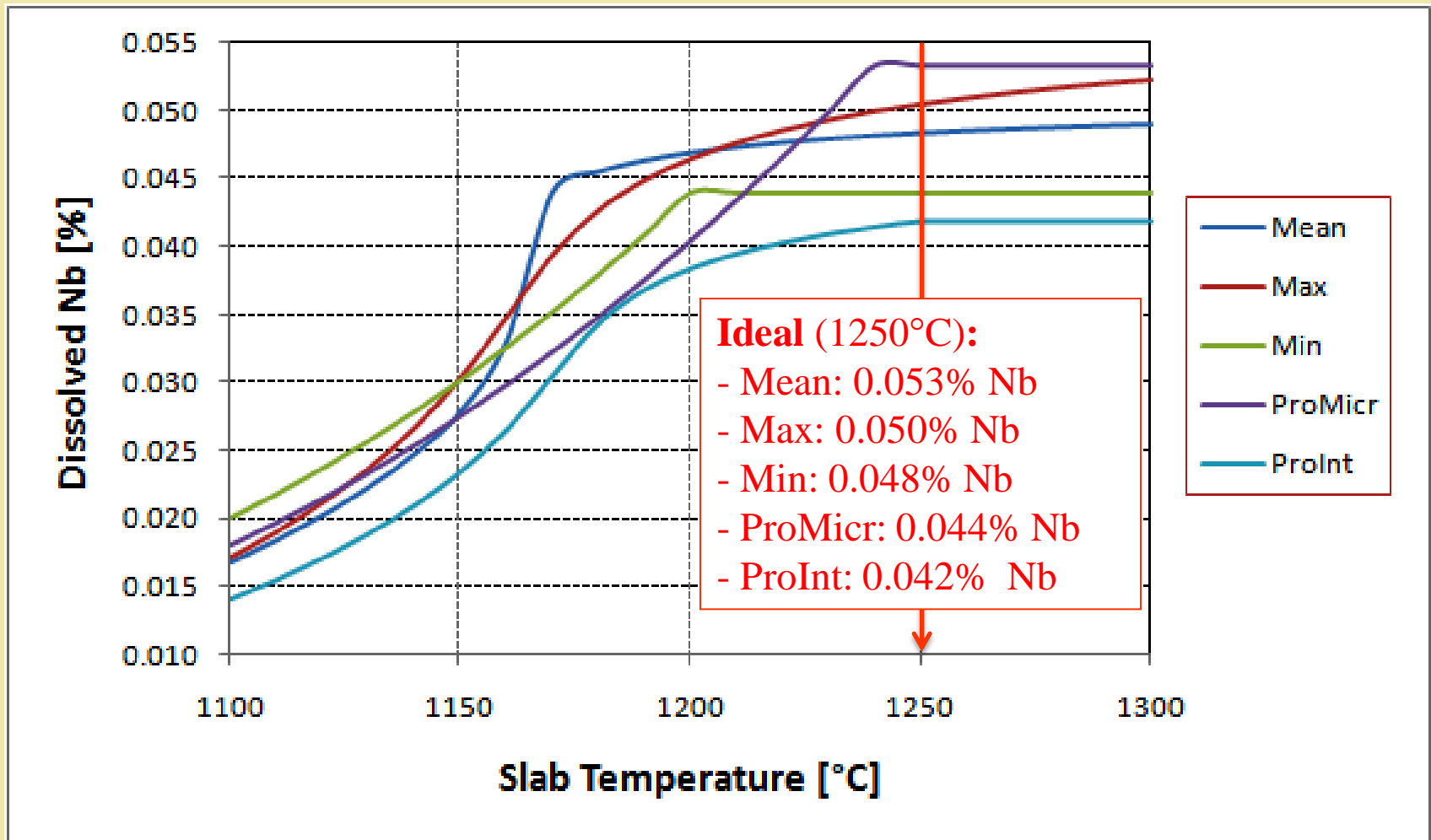
#2: ALLOY COMPOSITION

- **Solubilization evolution along temperature:**



#2: ALLOY COMPOSITION

- **Solubilization evolution along temperature:**



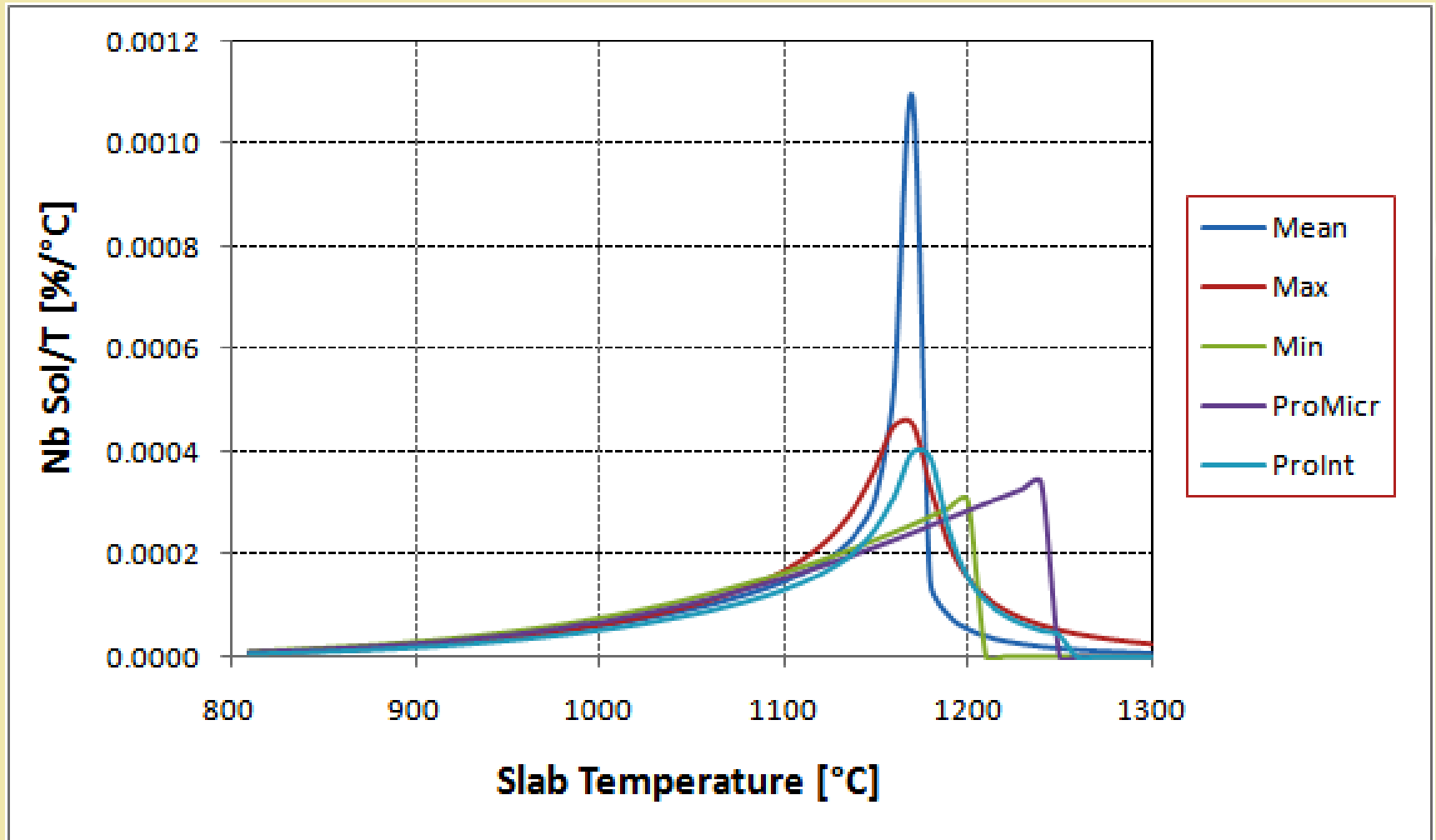
#2: ALLOY COMPOSITION



- **ProInt** alloy showed the worst solubilization evolution along temperature: **slow and limited**.
- **ProMed** alloy is the most convenient: **quick solubilization** at relatively low temperatures (**1150-1170°C**), followed by slight increase for higher temperatures.
- **Max** and **ProMicr** alloys showed **slower solubilization** along temperature, but higher than **ProMed** above 1200°C.
- **Min** alloy had quickest solubilization up to 1150°C and then was surpassed by all others but **ProInt**.

#2: ALLOY COMPOSITION

- **Solubilization rate evolution along temperature:**



#3: SKID MARKS + COMPOSITION



- As we have just shown, solute Nb levels across the slab depends on **local temperature** and steel **chemical composition**.
- As the slab is supported by water cooled skids during its travel through the reheating furnace, the slab portions near the skids have **significantly lower temperatures** than the remainder of the slab.
- According to real temperature measurements, while slab core temperature at the skid mark location is about **1193°C**, this same region, but away from the skids, is at **1225°C** - that is, a **32°C** difference.

#3: SKID MARKS + COMPOSITION



- Calculated **solubilized Nb** levels:

| Alloy | Nb_{sol} | | ΔNb_{sol} [%] | ΔNb_{sol} [%] |
|----------------|-------------------------|--------|---------------------------------|---------------------------------|
| | 1225°C | 1193°C | | |
| Mean | 0.048 | 0.046 | -0.002 | -4 |
| Min | 0.044 | 0.042 | -0.002 | -5 |
| Max | 0.049 | 0.045 | -0.004 | -8 |
| ProInt | 0.041 | 0.037 | -0.004 | -10 |
| ProMicr | 0.050 | 0.038 | -0.012 | -24 |

- It can be seen that **alloys with sluggish solubilization along temperature** tended to show **greater variations** of solute Nb levels across the slab.

#4: COMPOSITION + PRECIPITATE SIZE



- The **maximum size** of the precipitates that can be fully solubilized during slab reheating varies according to the steel **chemical composition**:
 - Mean: 250 nm
 - Max: 225 nm
 - ProMicr: 200 nm
 - Min: 175 nm
 - ProInt: 150 nm
- **Trend:** alloys with **lower and less balanced** levels of Nb, Ti, C and N tend to show **lower diffusion potential**, which makes more difficult the solubilization of their precipitates.

CONCLUSIONS



- Up to this moment, the understanding about the austenite solubilization kinetics in microalloyed steels is **somewhat incipient** due to the **experimental difficulties** associated with its study.
- However, solubilization control of microalloy elements during slab reheating is of **paramount importance** to the **consistency of the properties** of controlled rolled flat products.
- This work is a **small contribution** to get more knowledge about the **kinetics** of the austenite microalloy solubilization under industrial conditions.

CONCLUSIONS



- The **mathematical model** developed in this work highlighted the following main factors that affect **austenite microalloy solubilization**:
 - **Precise alloy chemical composition**, particularly C, N, Nb and Ti;
 - **Slab continuous casting conditions** that affect **Nb segregation** and, consequently, the **distribution and size of its precipitates**;
 - **Slab reheating conditions**, including the intensity of the so called **skid marks**.



***THANK YOU VERY MUCH
FOR YOUR ATTENTION!***